



Dynamics of Step Edges in Thin Film Growth

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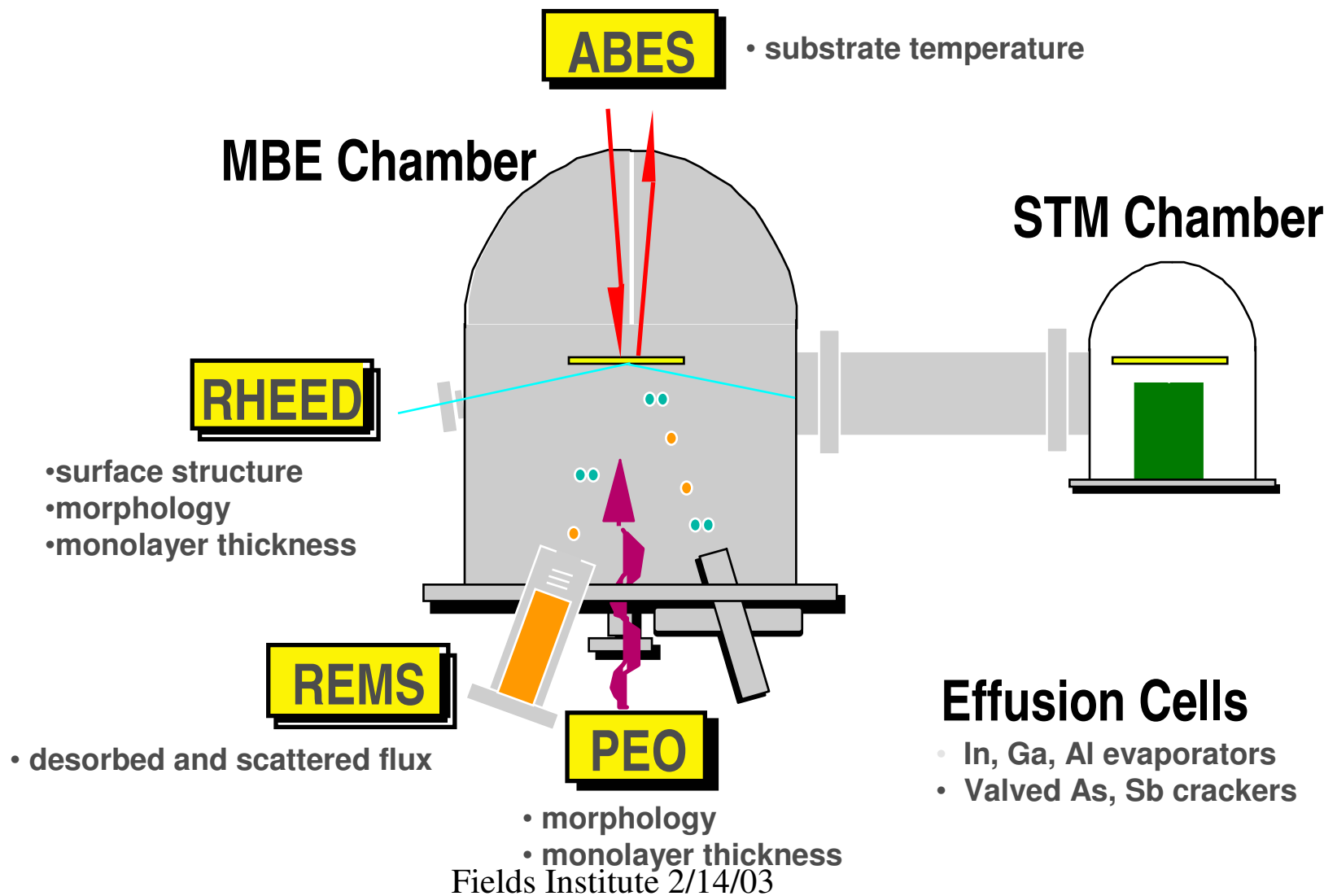
Outline



- Epitaxial Growth
 - molecular beam epitaxy (MBE)
 - Step edges and islands
- Mathematical models for epitaxial growth
 - atomistic: Solid-on-Solid using kinetic Monte Carlo
 - continuum: Villain equation
 - island dynamics: BCF theory
- Kinetic model for step edge
- Asymptotics
 - edge diffusion and line tension (Gibbs-Thomson) boundary conditions
- Conclusions



Growth and Analysis Facility at HRL

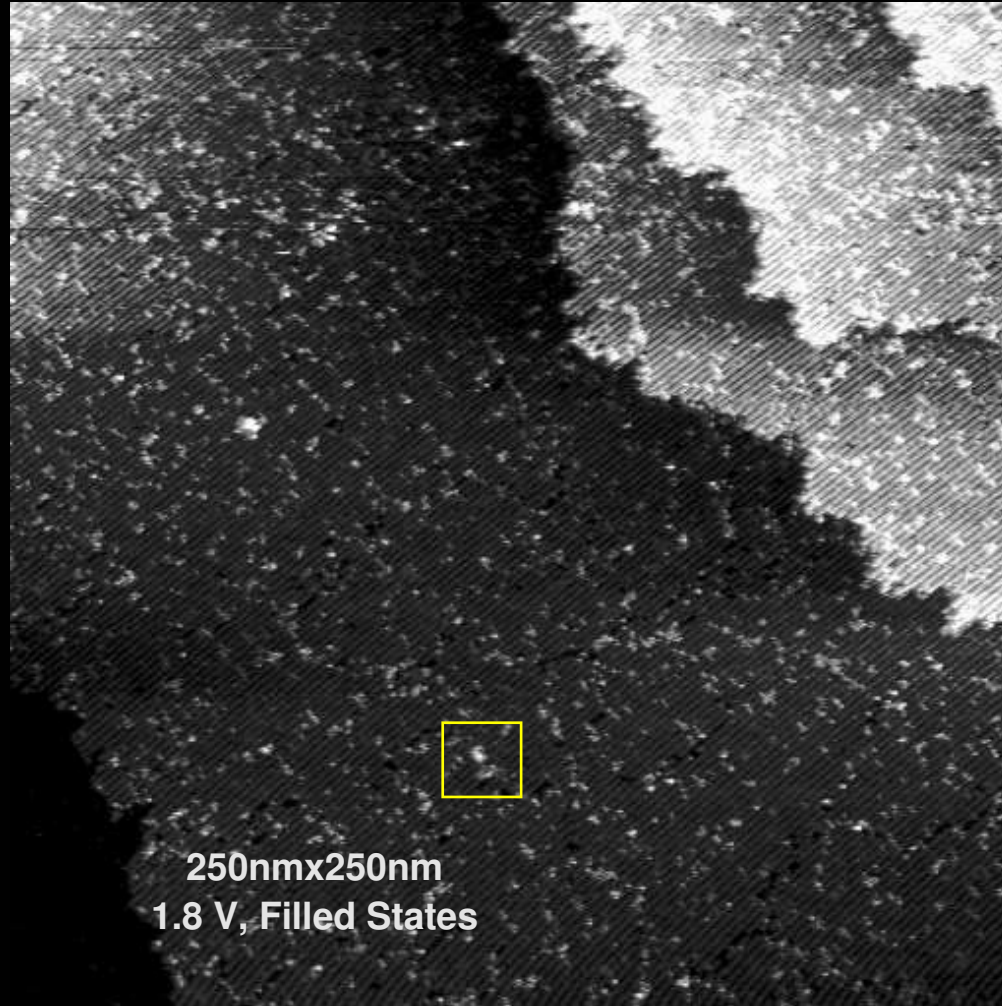




STM Image of InAs

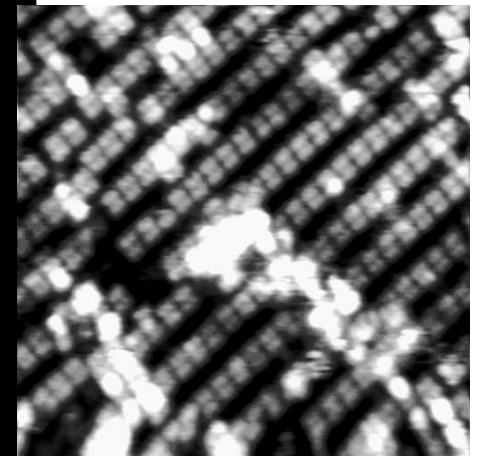


HRL whole-wafer STM
surface quenched from 450 °C, "low As"



250nmx250nm
1.8 V, Filled States

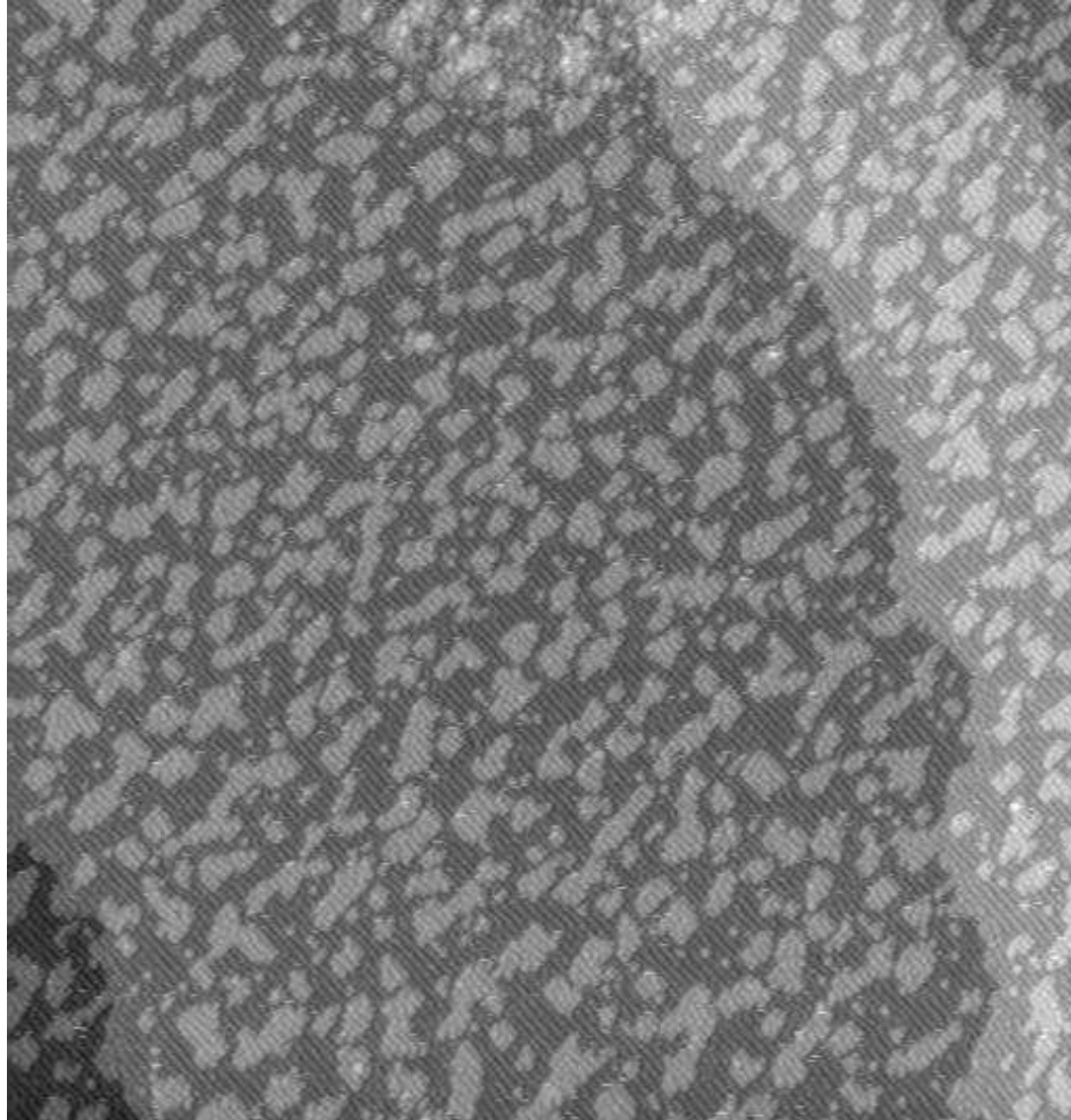
20nmx20nm



Barvosa-Carter,
Owen, Zinck
(HRL)



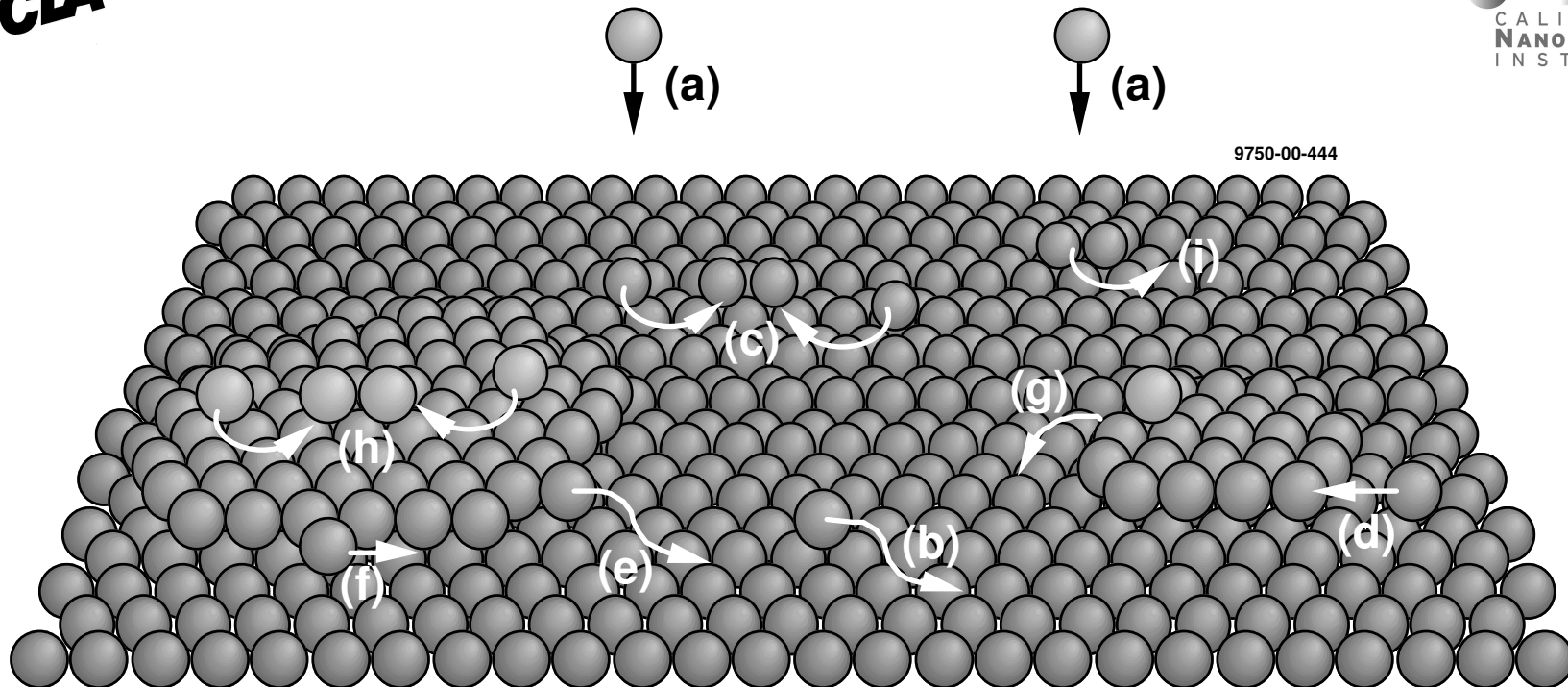
AlSb Growth by MBE



Barvosa-Carter and Whitman, NRL

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Basic Processes in Epitaxial Growth



- | | |
|----------------|----------------------------------|
| (a) deposition | (f) edge diffusion |
| (b) diffusion | (g) diffusion down step |
| (c) nucleation | (h) nucleation on top of islands |
| (d) attachment | (i) dimer diffusion |
| (e) detachment | |

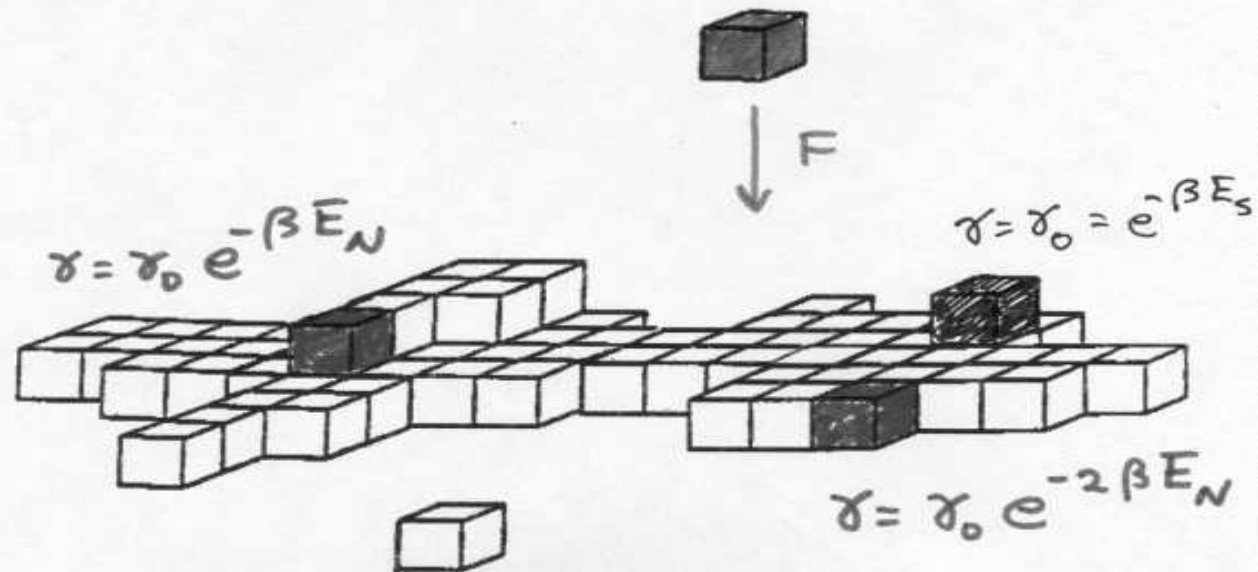


Solid-on-Solid Model



- Interacting particle system
 - Stack of particles above each lattice point
- Particles hop to neighboring points
 - random hopping times
 - hopping rate $D = D_0 \exp(-E/T)$,
 - E = energy barrier, depends on nearest neighbors
- Deposition of new particles
 - random position
 - arrival frequency from deposition rate
- Simulation using kinetic Monte Carlo method
 - Gilmer & Weeks (1979), Smilauer & Vvedensky, ...

Pair-bond solid-on-solid model



E_S : Substrate Bond Energy

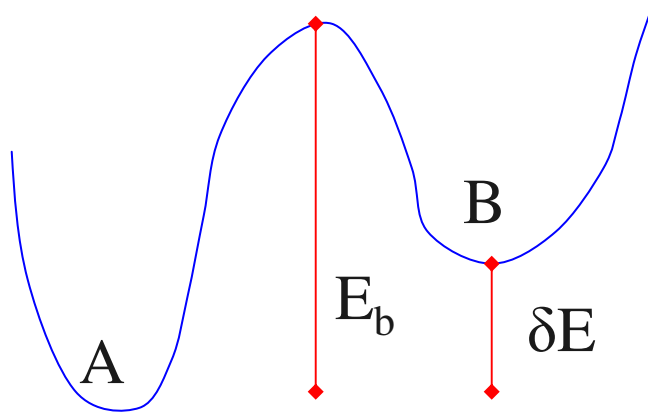
E_N : Nearest Neighbor Bond Energy



Kinetic Monte Carlo

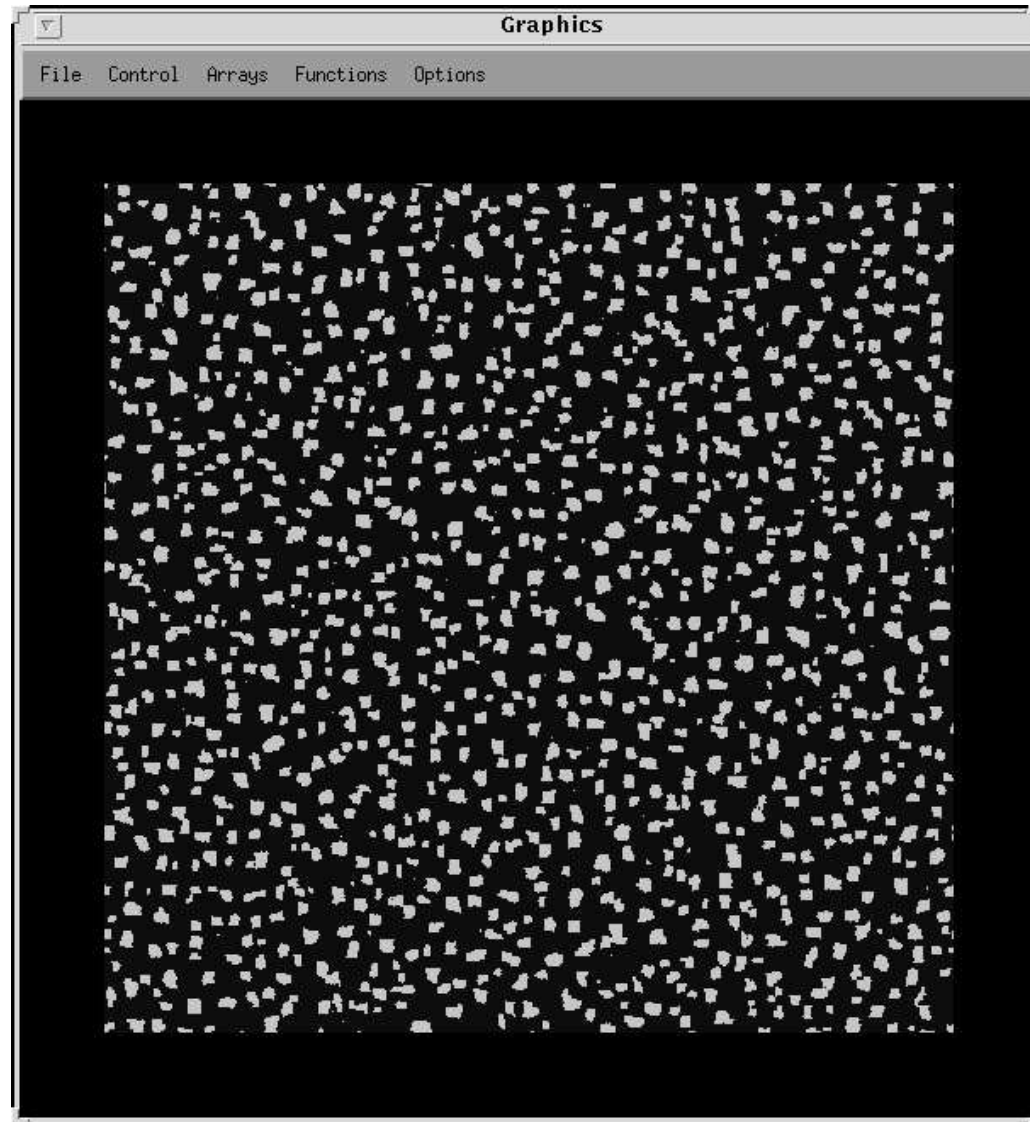


- Random hopping from site $A \rightarrow B$
- hopping rate $D_0 \exp(-E/T)$,
 - $E = E_b$ = energy barrier between sites
 - not δE = energy difference between sites



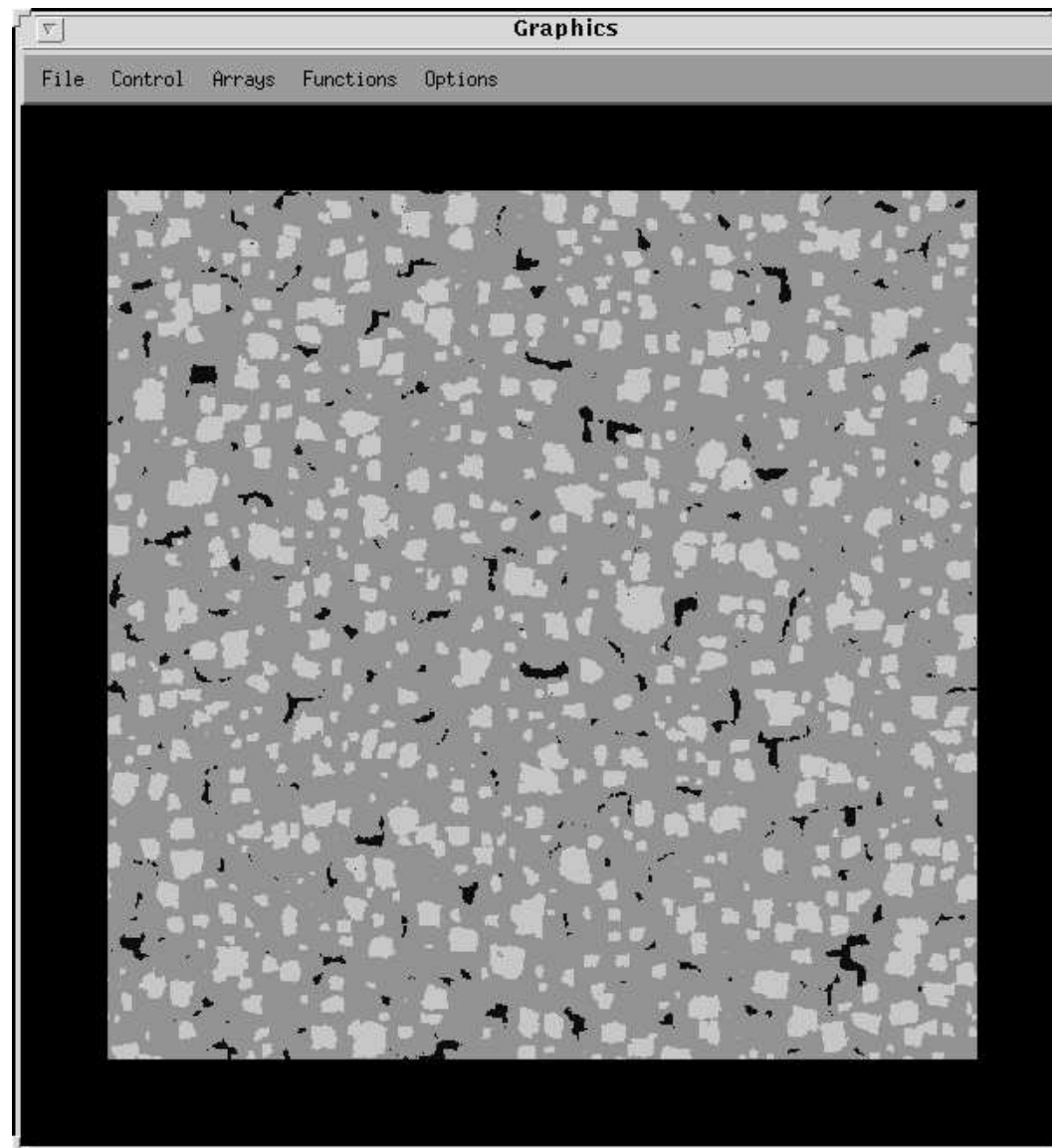


SOS Simulation for coverage=.2



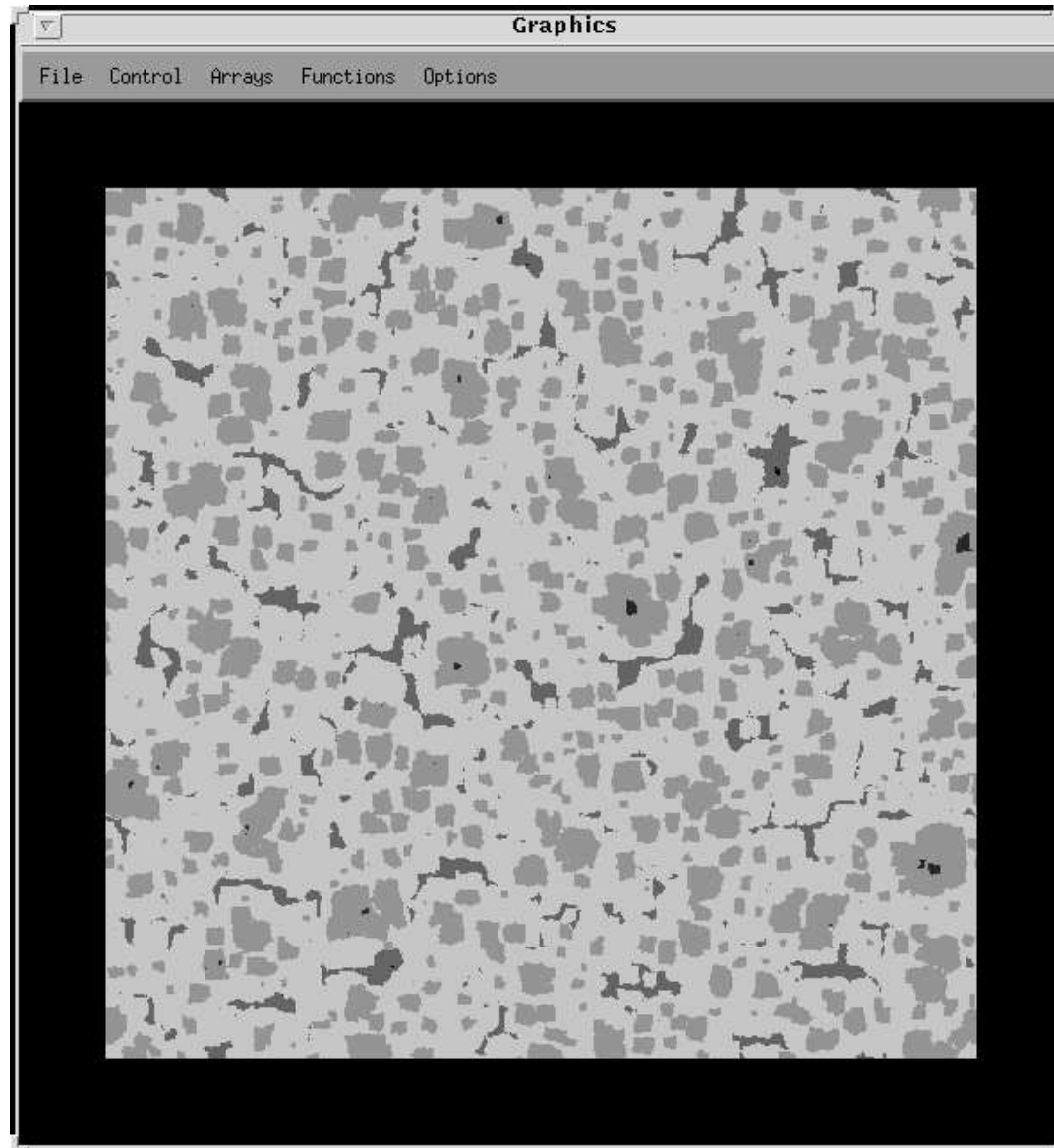


SOS Simulation for coverage=10.2



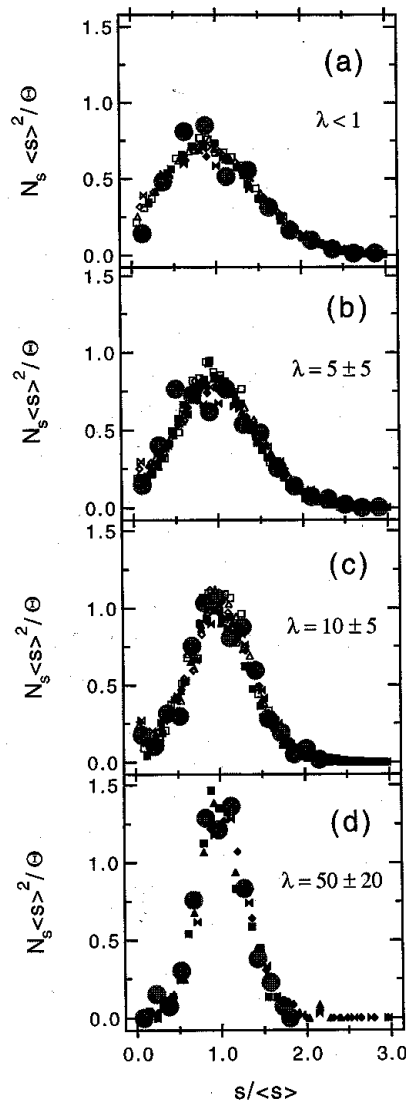


SOS Simulation for coverage=30.2

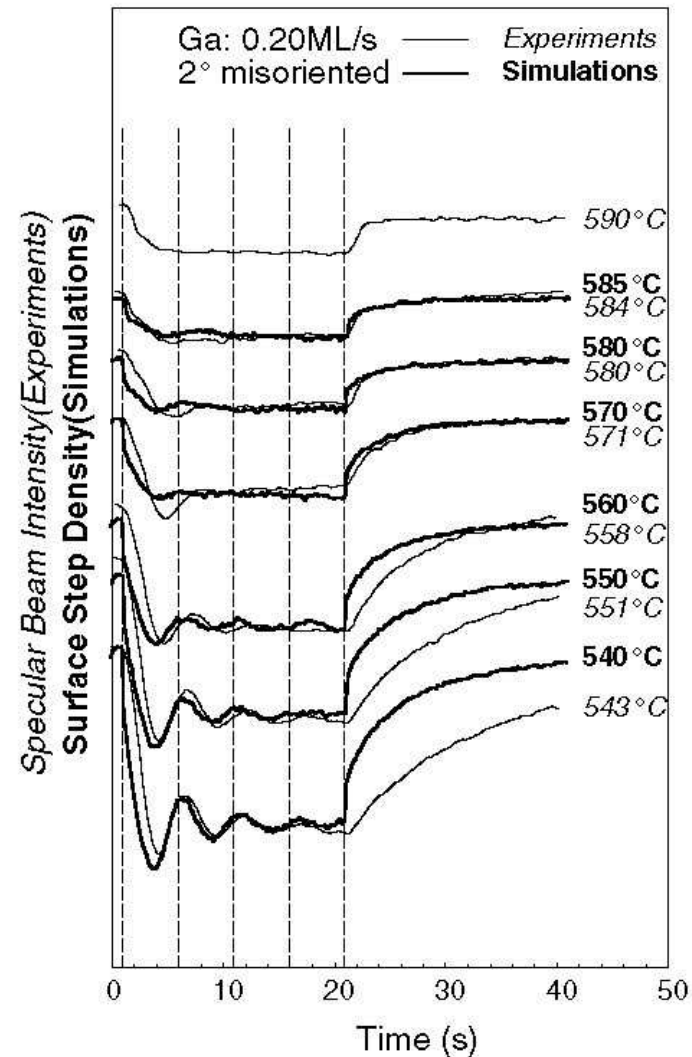




Validation of SOS Model: Comparison of Experiment and KMC Simulation (Vvedensky & Smilauer)



Island size density



Step Edge Density (RHEED)



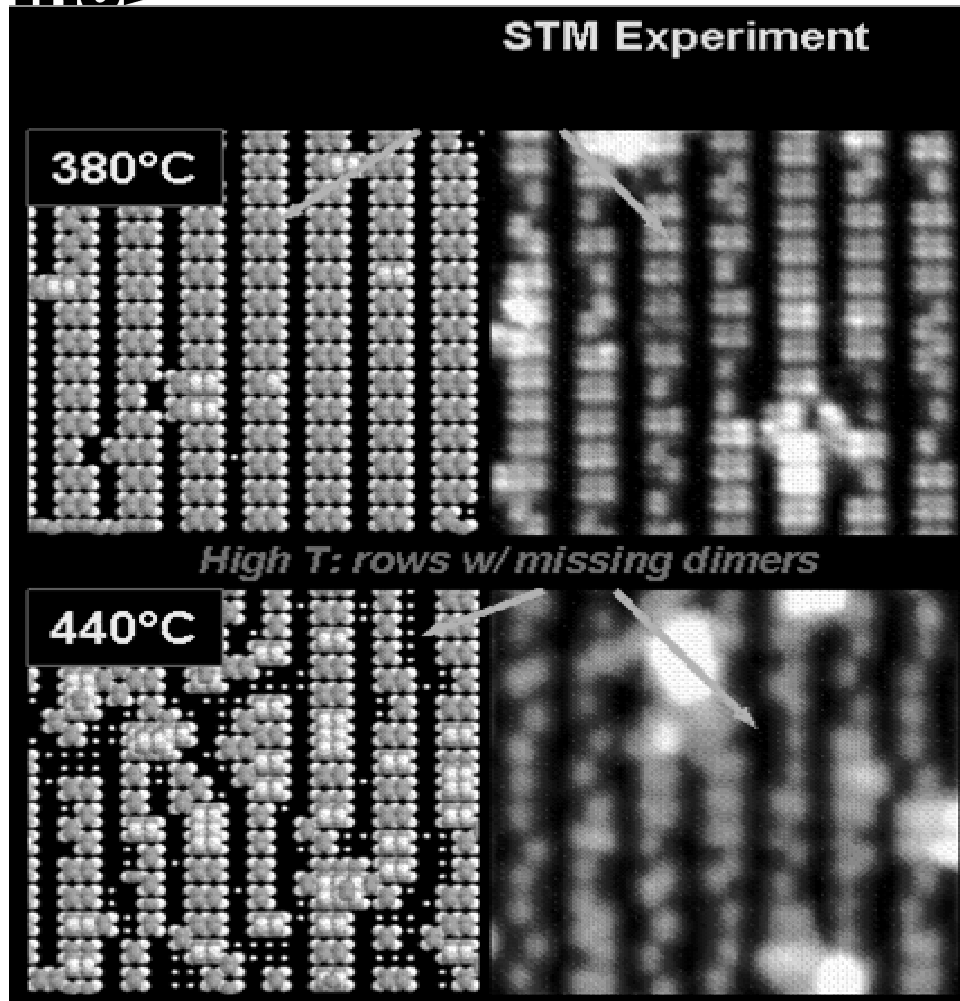
Difficulties with SOS/KMC



- Difficult to analyze
- Computationally slow
 - adatom hopping rate must be resolved
 - difficult to include additional physics, e.g. strain
- Rates are empirical
 - idealized geometry of cubic SOS
 - cf. “high resolution” KMC



High Resolution KMC Simulations



- InAs
- zinc-blende lattice, dimers
- rates from ab initio computations
- computationally intensive
 - many processes
- describes dynamical info (cf. STM)
- similar work
 - Vvedensky (Imperial)
 - Kratzer (FHI)

High resolution KMC (left); STM images (right)

Gyure, Barvosa-Carter (HRL), Grosse (UCLA, HRL)

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Continuum Theory for Epitaxial Growth



- Villain equation (1991) $h(x,y,t)$ = height

$$h_t = \underbrace{-\Delta^2 h}_{\text{surface diffusion}} + \underbrace{\Delta h}_{\text{desorption}} + \underbrace{\Delta(|\nabla h|^2)}_{\text{nonlinearity}} + \underbrace{F}_{\text{mean deposition}} + \underbrace{\eta}_{\text{deposition noise}}$$

- Related work: Ortiz; Kohn; ...
- Describes rough growth
 - inapplicable to morphology of very thin layers ($h=h(t)$)
- Range of validity is uncertain
 - incomplete derivation (dynamic vs. thermodynamic)
 - surface diffusion: $E[h] = \int \kappa^2 ds$, no atomistic derivation



Island Dynamics



- Burton, Cabrera, Frank (1951)
- Epitaxial surface
 - adatom density ρ
 - continuum in lateral direction, atomistic in growth direction
- Adatom diffusion equation, equilibrium BC, step edge velocity

$$\rho_t = D \Delta \rho + F$$

$$\rho = \rho_{eq}$$

$$v = D [\partial \rho / \partial n]$$

- Line tension (Gibbs-Thomson) in BC and velocity

$$D \partial \rho / \partial n = c(\rho - \rho_{eq}) + c \kappa$$

$$v = D [\partial \rho / \partial n] + c \kappa_{ss}$$

- similar to surface diffusion, since $\kappa_{ss} \sim \chi_{ssss}$

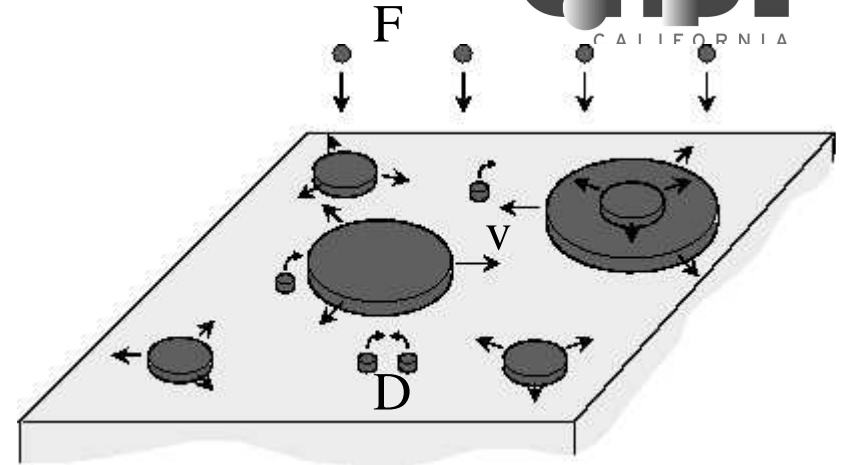


Island Dynamics/Level Set Equations



- Variables
 - N = number density of islands
 - Γ_k = island boundaries of height k represented by “level set function” ω

$$\Gamma_k(t) = \{x : \omega(x,t) = k\}$$
 - adatom density $\theta(x,y,t)$



- Adatom diffusion equation

$$\rho_t - D \Delta \rho = F - dN/dt$$

- Island nucleation rate

$$dN/dt = \int D \sigma_1 \rho^2 dx$$

σ_1 = capture number for nucleation

- Level set equation (motion of Γ)

$$\phi_t + v \text{ grad } \phi = 0$$

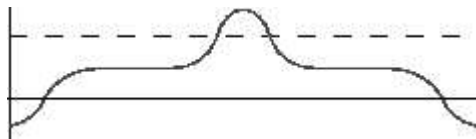
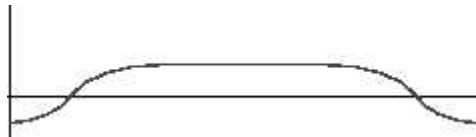
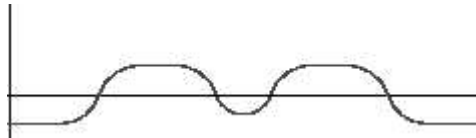
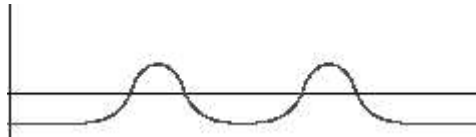
v = normal velocity of boundary Γ



The Levelset Method



Level Set Function ϕ

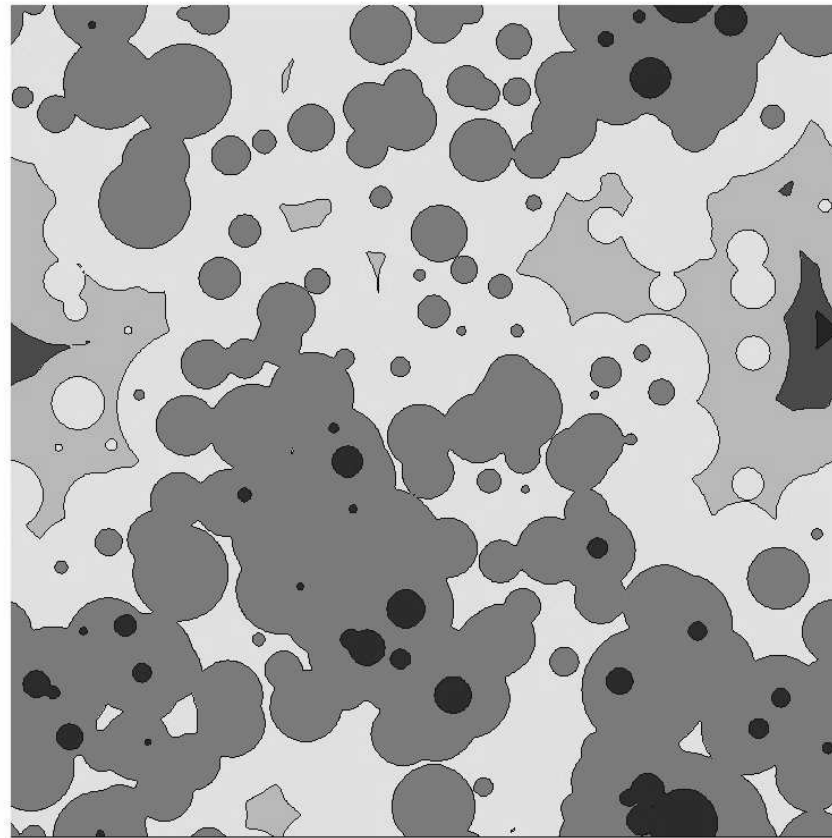


Surface Morphology



Level Contours after 40 layers

In the multilayer regime, the level set method produces results that are qualitatively similar to KMC methods.

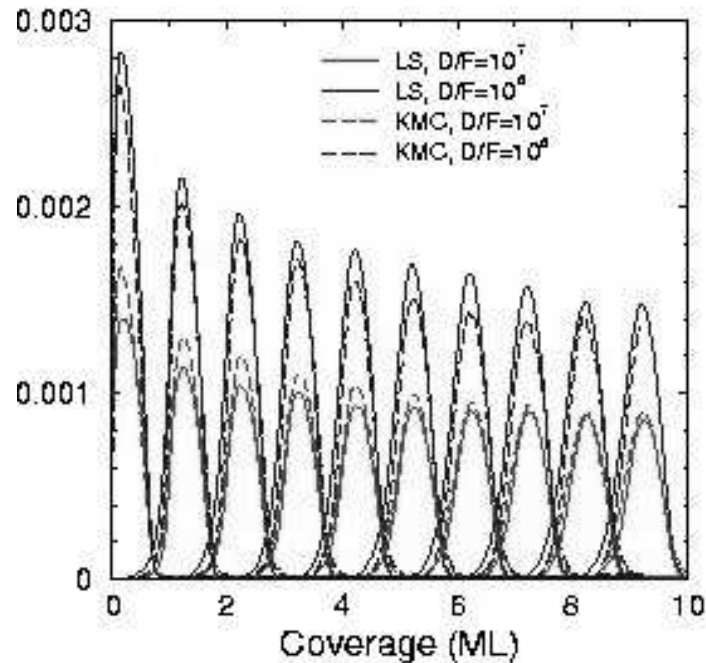




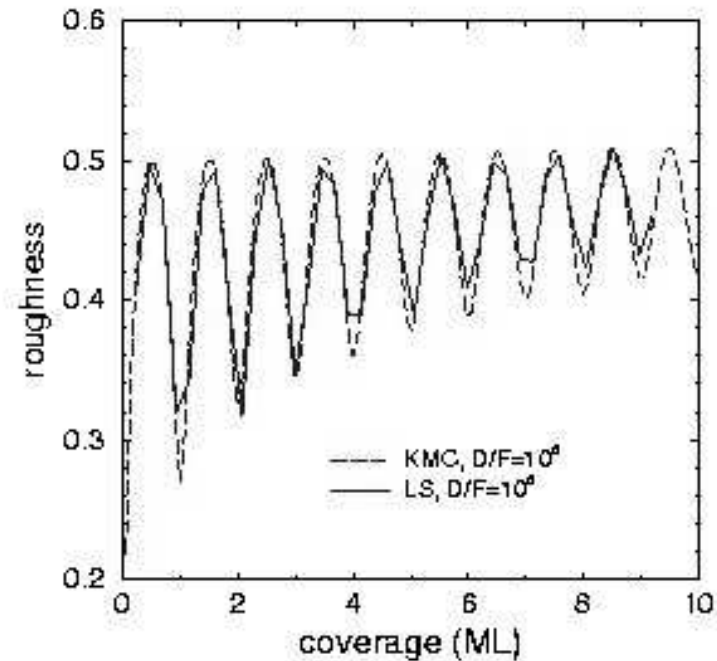
Multilayer Comparison Levelset - KMC



Island Densities



Surface Roughness



We choose edge diffusion in KMC as $D_{\text{edge}}/D = 0.01$.

LS = level set implementation of island dynamics

UCLA/HRL/Imperial group,

Chopp, Smereka

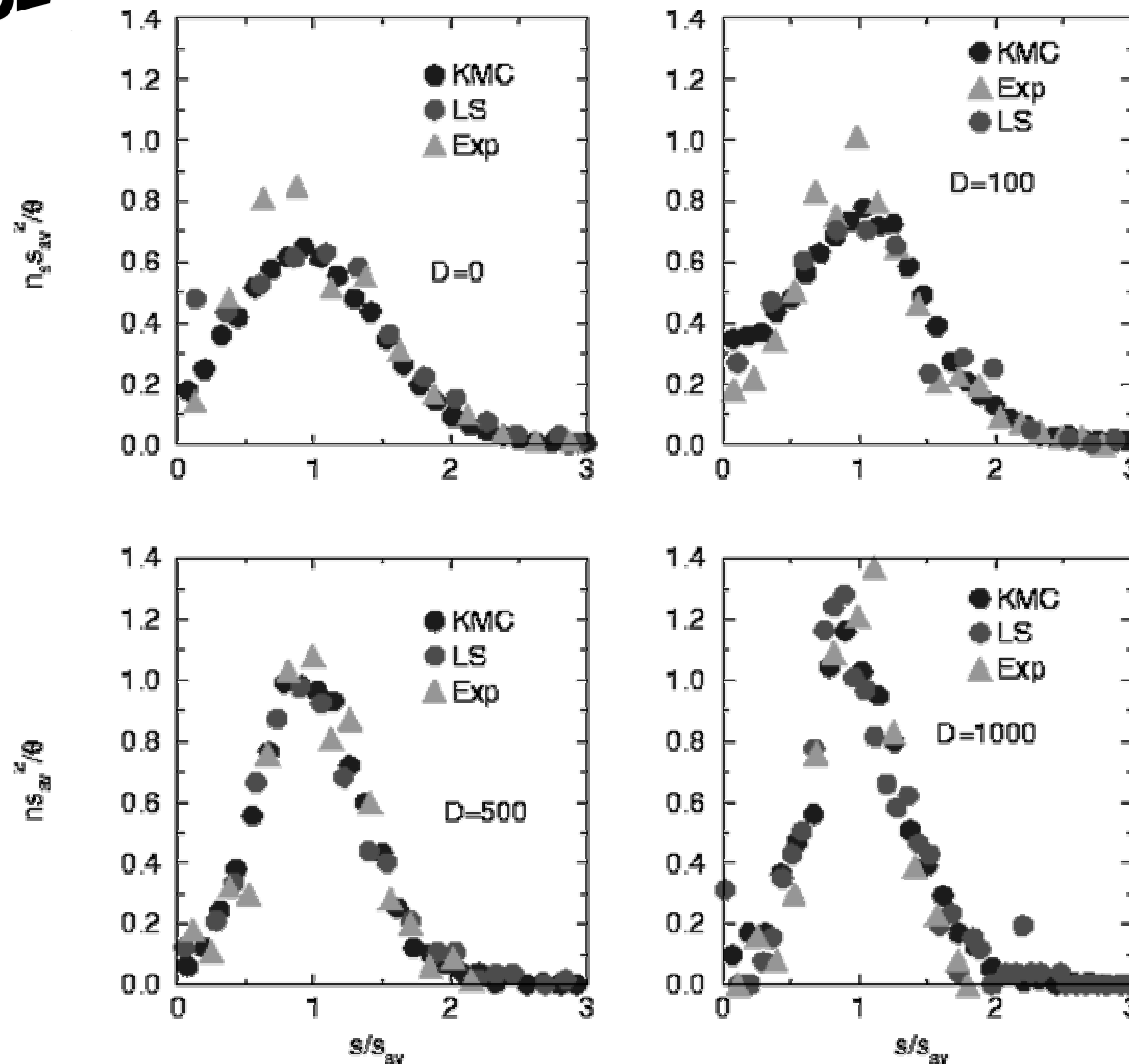
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Island size distributions



Experimental Data for
Fe/Fe(001),
Strosio and Pierce,
Phys. Rev. B 49 (1994)

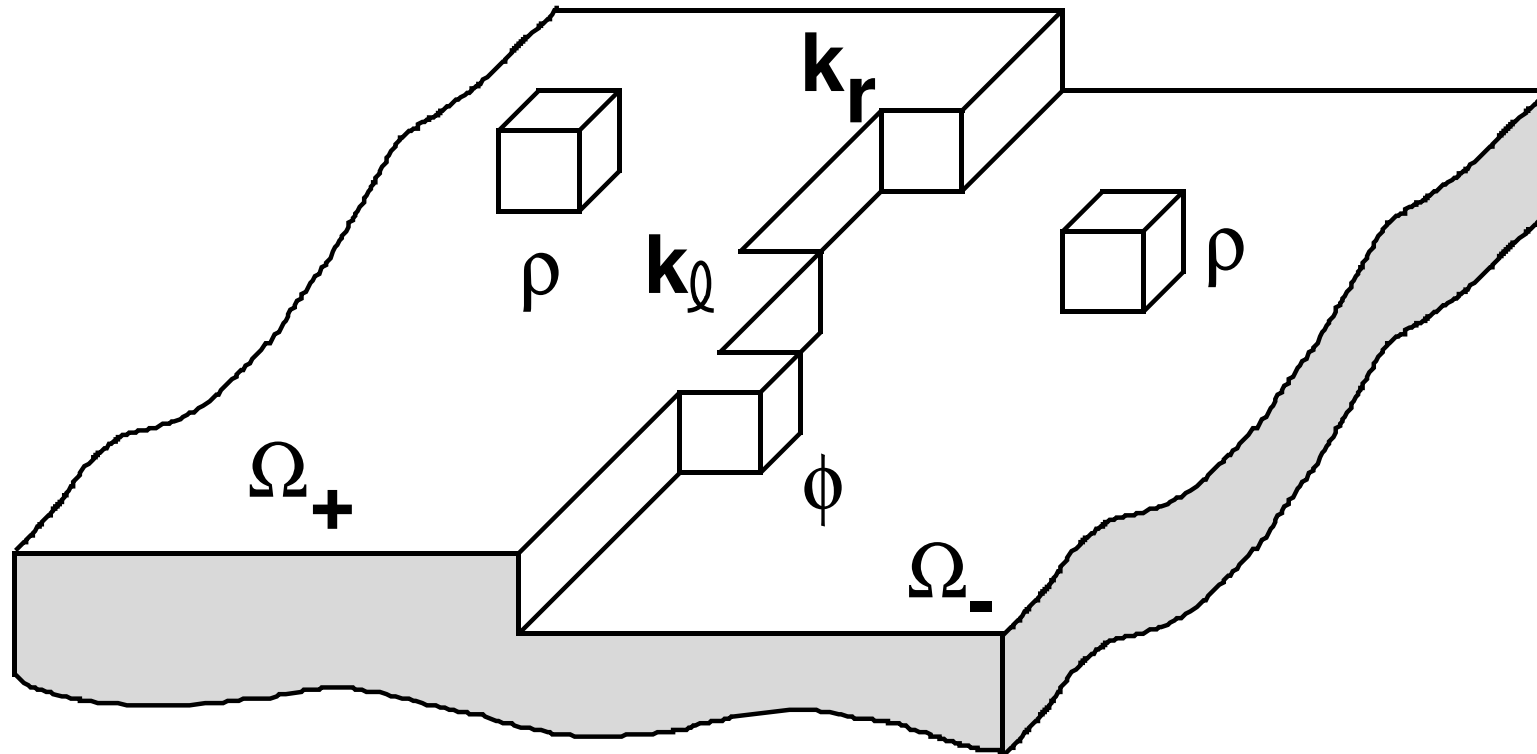


Stochastic
nucleation and
breakup of
islands



Kinetic Theory for Step Edge Dynamics and Adatom Boundary Conditions

Step Edge Components



- adatom density θ
- edge atom density ρ
- kink density (left, right) k
- terraces (upper and lower) Ω



Diffusion Coefficients



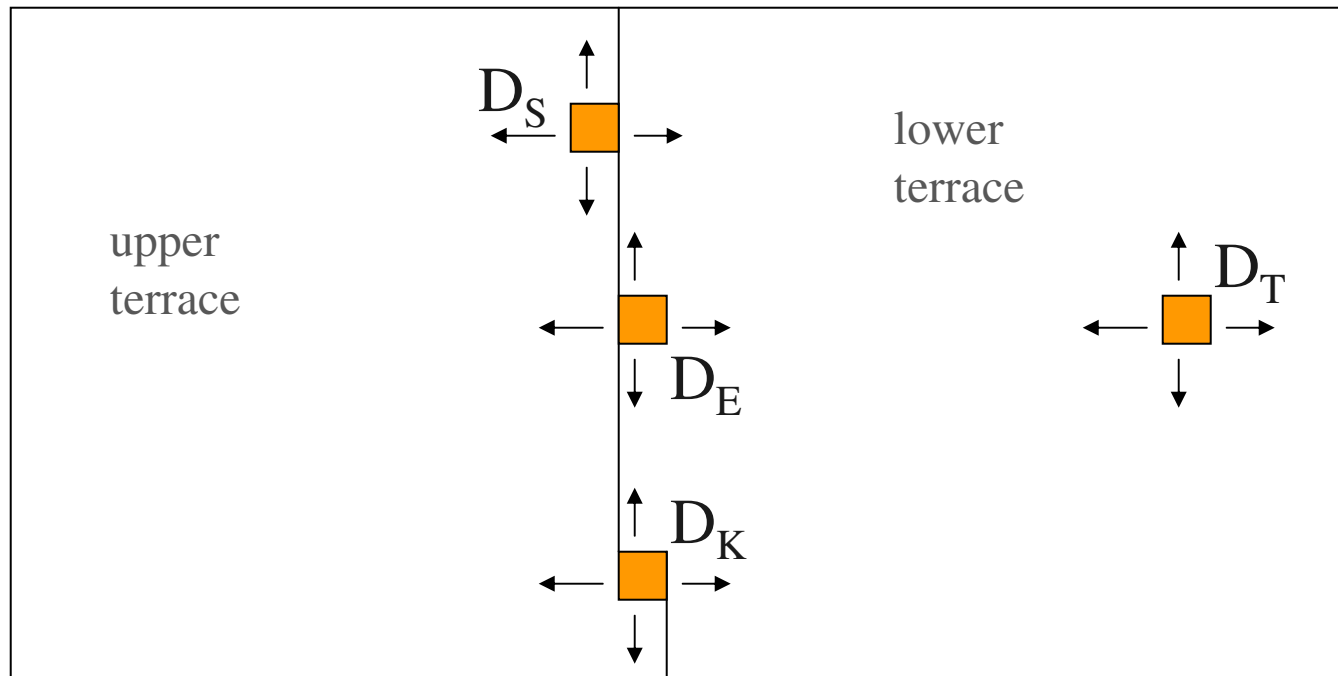
Hopping rate = diffusion coefficient, for bond energy E

D_T hopping rate on terrace

$D_E = D_T e^{-E/T}$ hopping rate along and off of edge

$D_K = D_T e^{-2E/T}$ hopping rate from kink

$D_S = D_T e^{-3E/T}$ hopping rate out of a uniform edge





BCF Theory



- Equilibrium of step edge with terrace
- Gibbs distributions

$$\rho = e^{-2E/T}$$

$$\varphi = e^{-E/T}$$

$$k = 2e^{-E/2T}$$

- Derivation from detailed balance
- BCF includes kinks of multi-heights

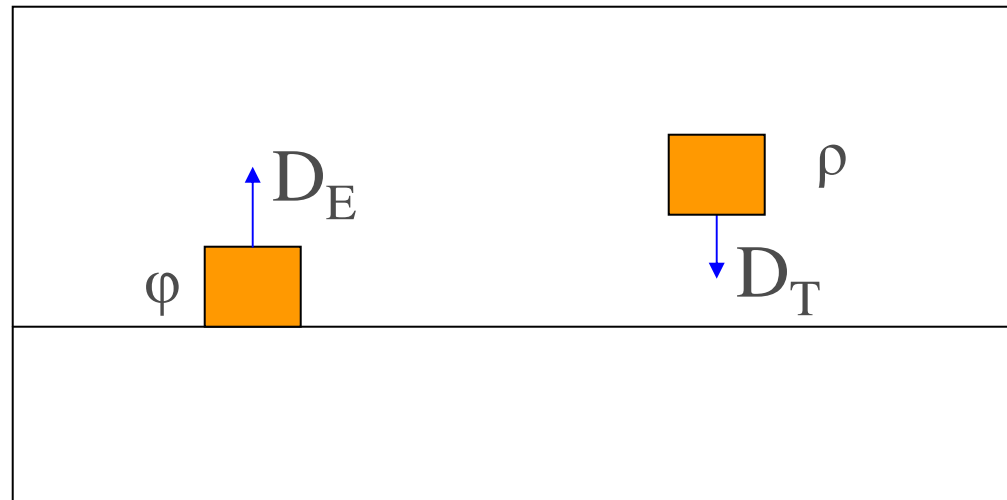


Detailed Balance: Attachment/Detachment at Edge



edge atom \leftrightarrow terrace adatom:

$$D_E \varphi = D_T \rho$$



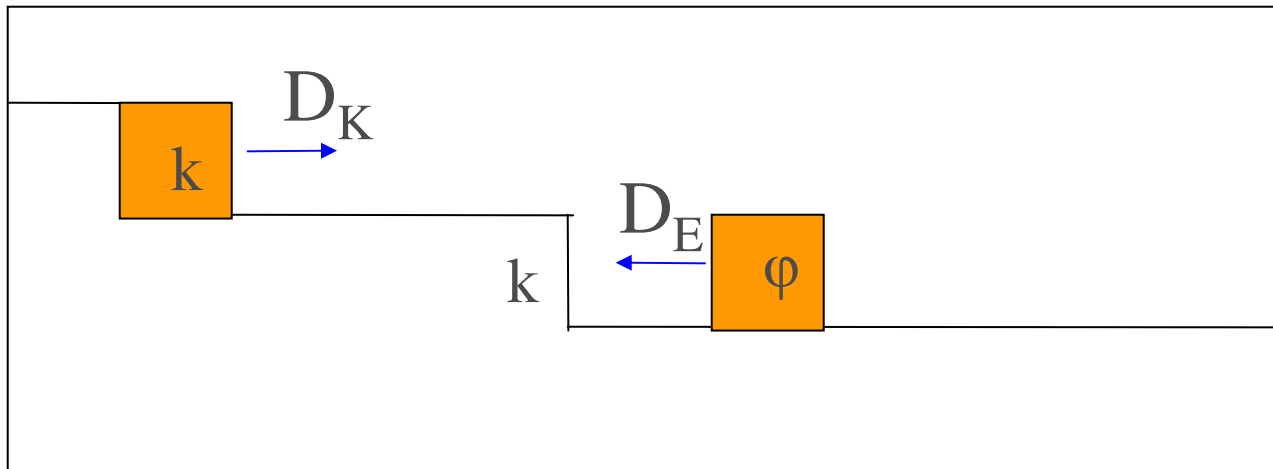


Detailed Balance: Attachment/Detachment at Kinks



kink \leftrightarrow edge atom:

$$D_K k = D_E k \varphi$$



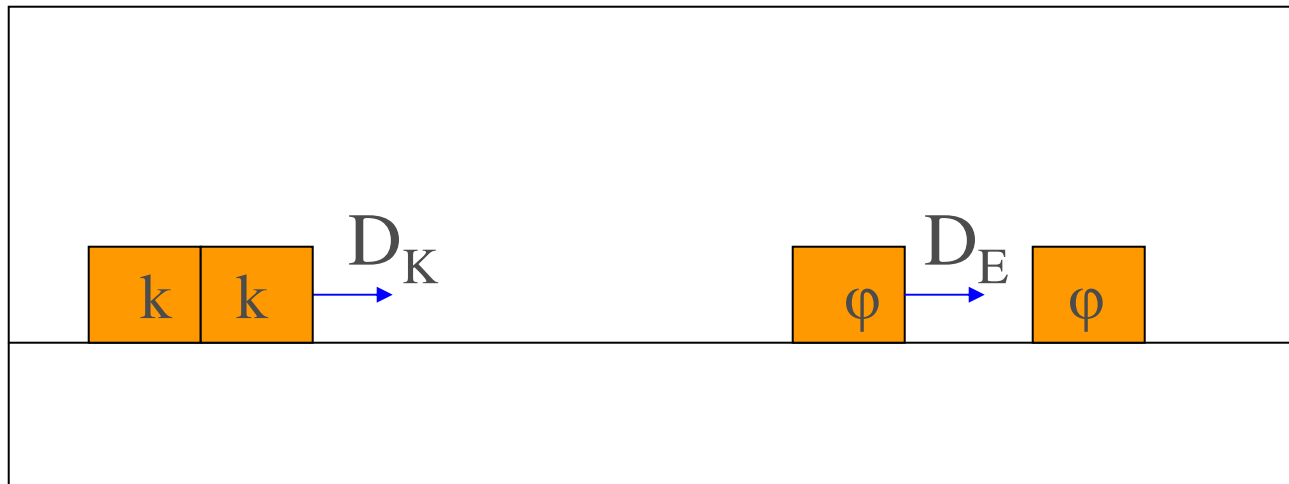


Detailed Balance: Nucleation/Breakup of Kink Pairs



kink pair (“island”) \leftrightarrow edge atom pair

$$D_K (1/4) k^2 = D_E \phi^2$$





Detailed Balance:

Creation/Filling of Holes along Edge

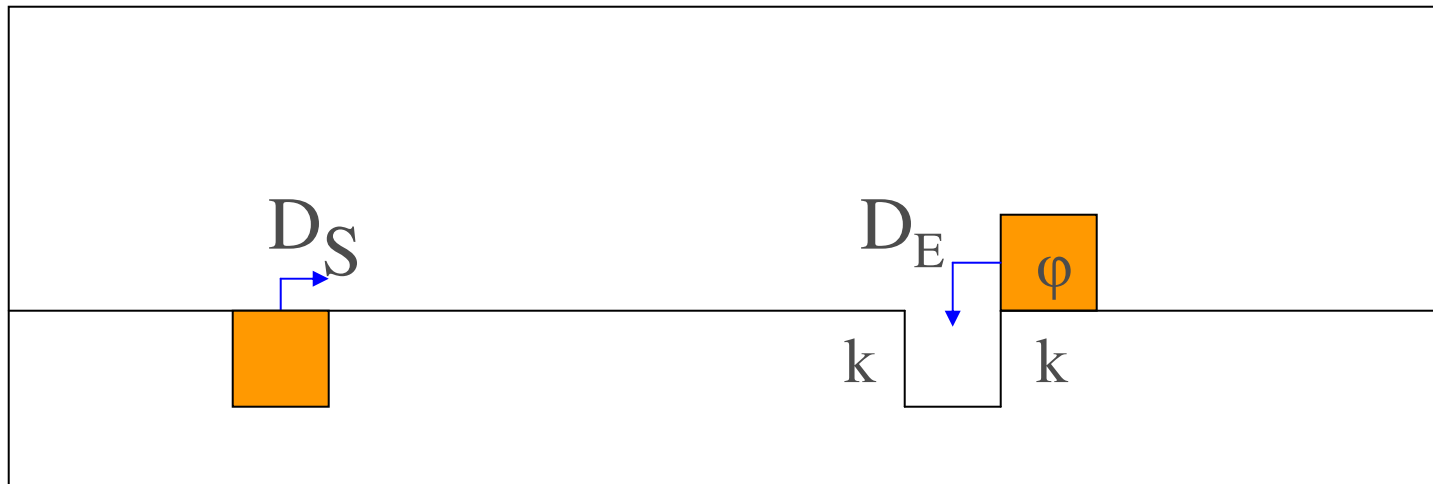


kink pair (“hole”) + edge atom

↔ straight step:

$$D_E (1/4) k^2 \phi = D_S$$

Filling in hole is key step: completion of a row





Detailed Balance



edge atom \leftrightarrow terrace adatom:

$$D_E \varphi = D_T \rho$$

kink \leftrightarrow edge atom:

$$D_K k = D_E k \varphi$$

kink pair ("island")

\leftrightarrow edge atom pair:

$$D_K (1/4) k^2 = D_E \varphi^2$$

kink pair ("hole") + edge atom

\leftrightarrow straight step:

$$D_E (1/4) k^2 \varphi = D_S$$

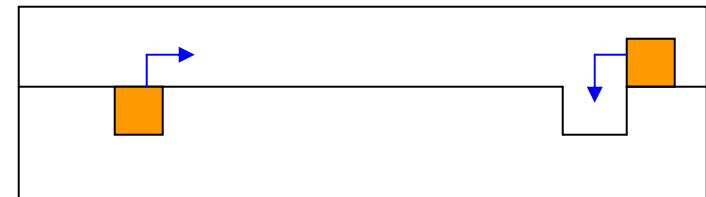
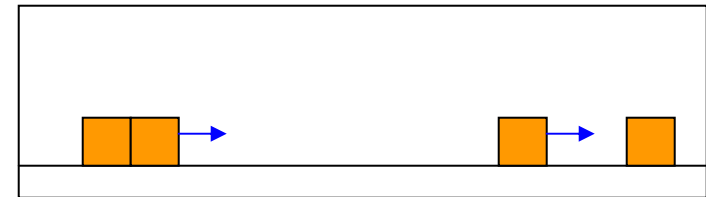
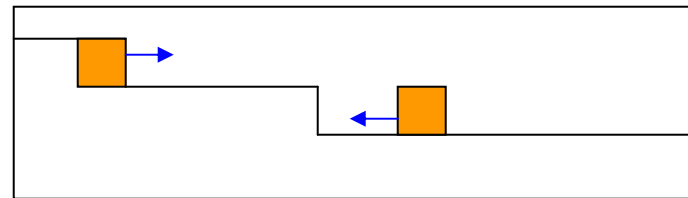
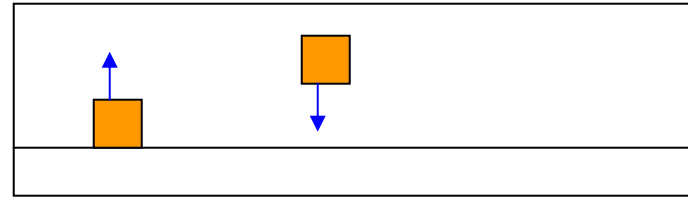
Conclusions:

$$\rho = D_K / D_T = e^{-2E/T}$$

$$\varphi = D_K / D_E = e^{-E/T}$$

$$k = 2(D_S / D_K)^{1/2} = 2e^{-E/2T}$$

$$D_S D_E = D_K^2$$



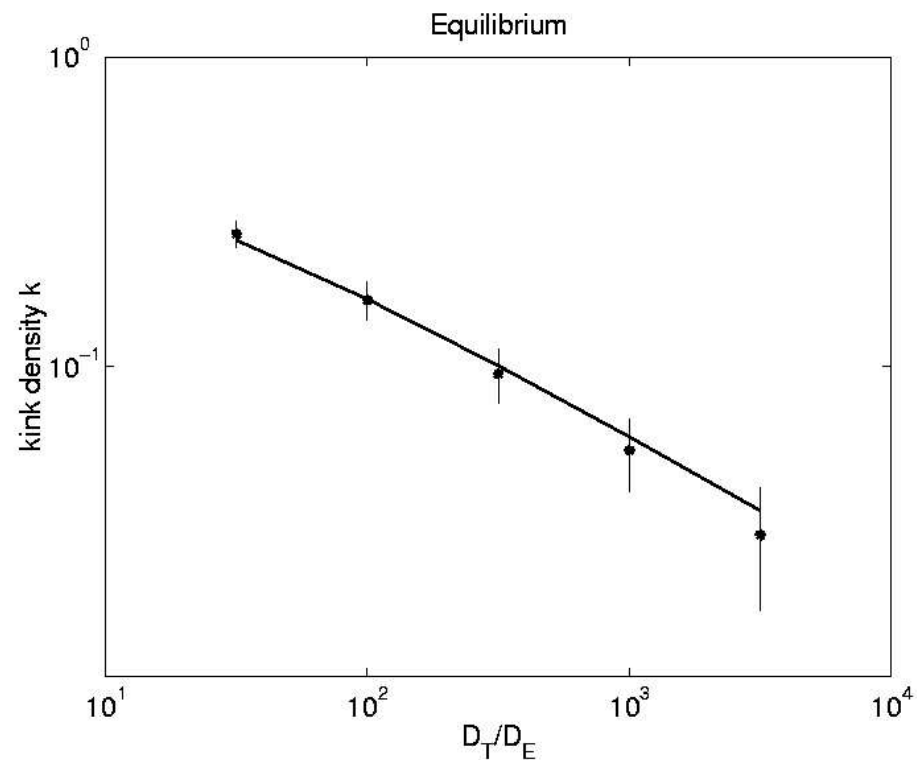
Equilibrium Solution

$$\rho = (D_E/D_T)\varphi$$

$$\varphi = k^2/4$$

$$k = 2\sqrt{D_K/D_E}$$

- Solution for $F=0$ (no growth)
- Same as BCF theory
- D_T , D_E , D_K are diffusion coefficients (hopping rates) on Terrace, Edge, Kink in SOS model



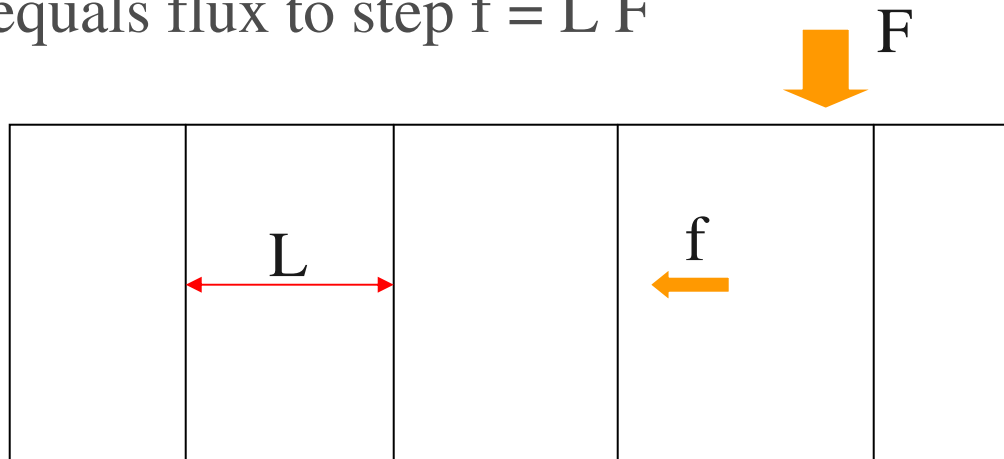
Comparison of results from theory(-)
and KMC/SOS (●)



Kinetic Steady State



- Deposition flux F
- Vicinal surface with terrace width L
- No detachment from kinks or step edges, on growth time scale
 - detailed balance not possible
- Advance of steps is due to attachment at kinks
 - equals flux to step $f = L F$



Detailed Balance

edge atom \leftrightarrow terrace adatom:

$$D_E \phi = D_T \rho$$

kink \leftrightarrow edge atom:

$$D_K k = D_E k \phi$$

kink pair ("island")

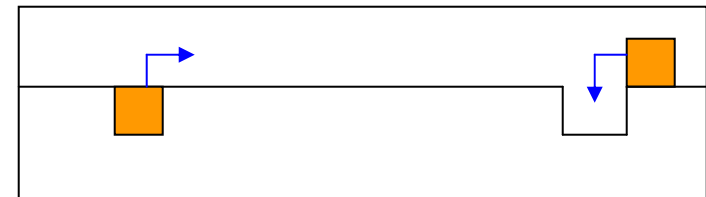
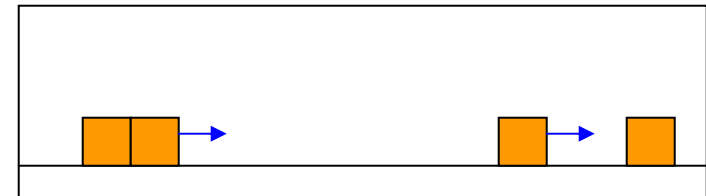
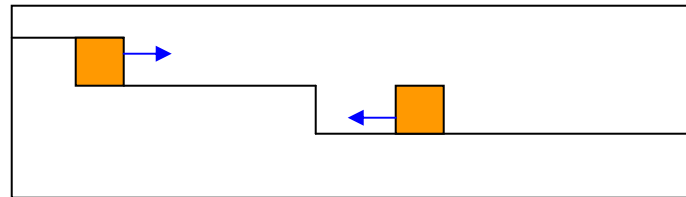
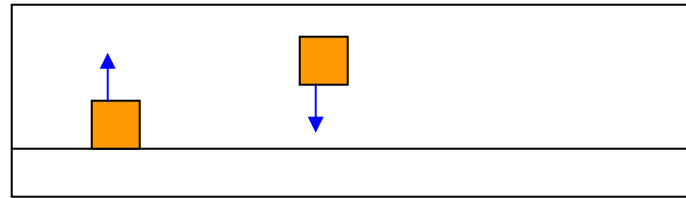
\leftrightarrow edge atom pair:

$$D_K (1/4) k^2 = D_E \phi^2$$

kink pair ("hole") + edge atom

\leftrightarrow straight step:

$$D_S = D_E (1/4) k^2 \phi$$





Kinetic

~~Detailed~~ Balance



edge atom \leftrightarrow terrace adatom:

$$D_E \phi = D_T \rho$$

kink \leftrightarrow edge atom:

$$\cancel{D_K k} = D_E k \phi$$

kink pair ("island")

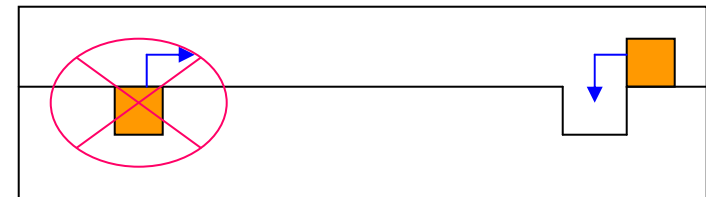
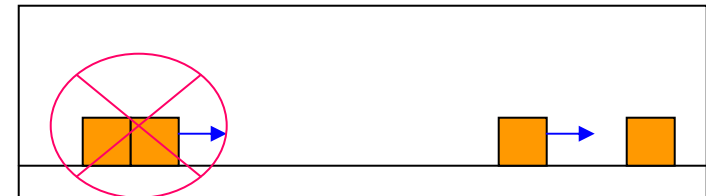
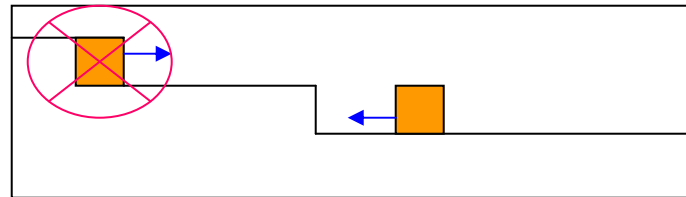
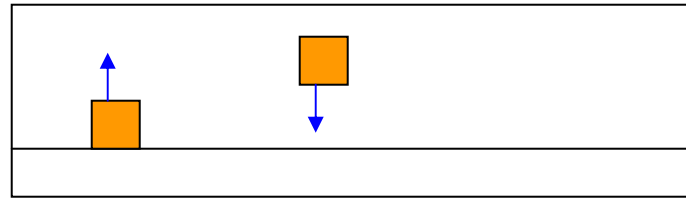
\leftrightarrow edge atom pair:

$$\cancel{D_K (1/4) k^2} = D_E \phi^2$$

kink pair ("hole") + edge atom

\leftrightarrow straight step:

$$\cancel{D_S} = D_E (1/4) k^2 \phi$$



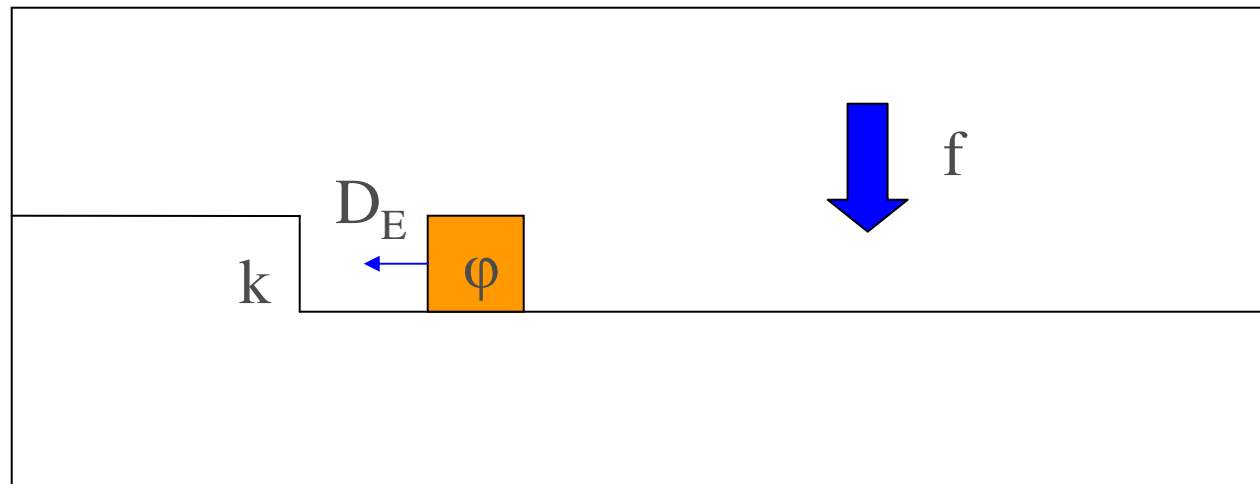


Kinetic Steady State: Flux to Step = Flux to Kinks



flux to kinks \leftrightarrow flux to edge :

$$D_E k \varphi = f = L F$$





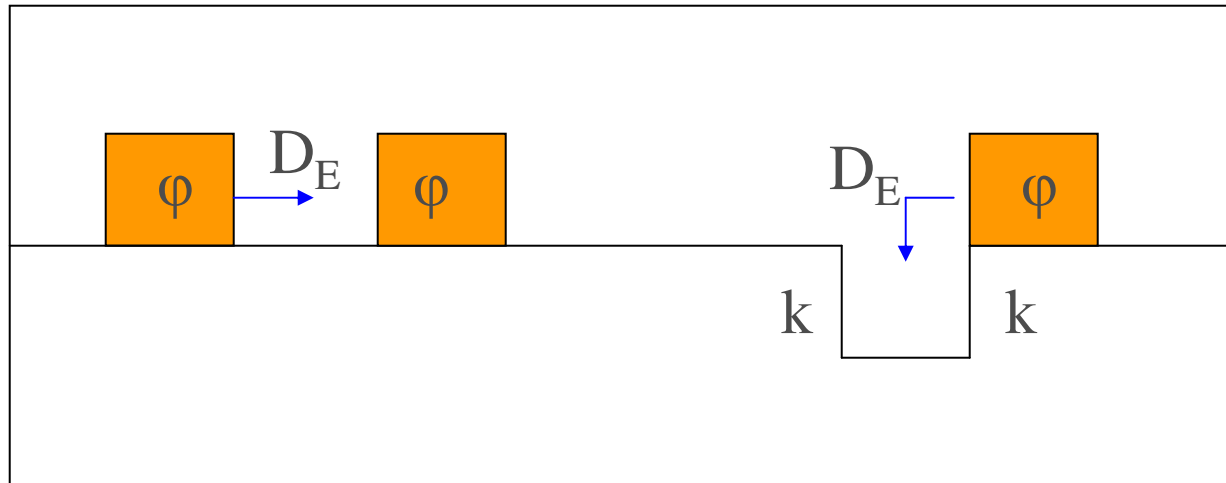
Kinetic Steady State: Kink Nucleation Rate = Hole Fill-in Rate



creation of kink pairs (“island”)

↔ filling in holes:

$$D_E \phi^2 = D_E (1/4) k^2 \phi$$





Kinetic Steady State



edge atom \leftrightarrow terrace adatom:

$$D_E \varphi = D_T \rho$$

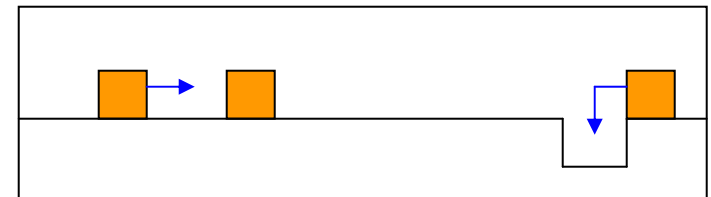
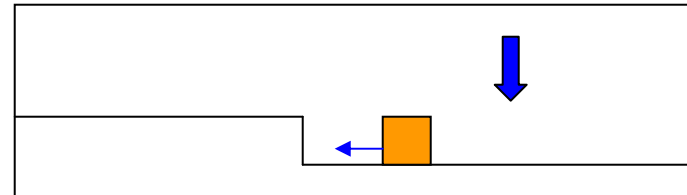
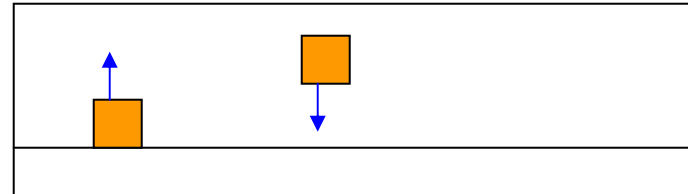
flux to kinks \leftrightarrow flux to edge :

$$D_E k \varphi = f = L F$$

creation of kink pairs ("island")

\leftrightarrow filling in holes:

$$D_E \varphi^2 = D_E k^2 \varphi$$



Conclusions:

$$\rho = (D_E / D_T) \varphi$$

$$\varphi = k^2$$

$$k = (L F / D_E)^{1/3}$$

F= deposition flux, L= terrace width



Kinetic Steady State

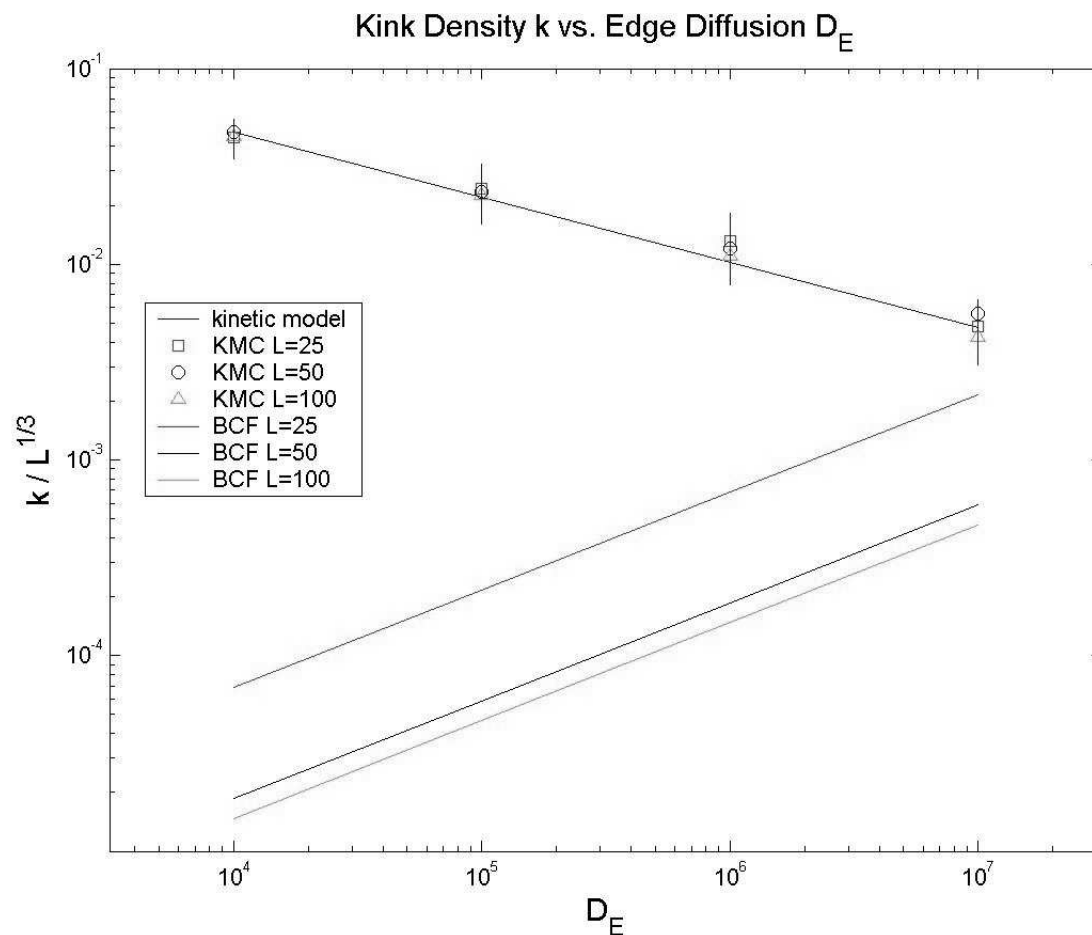


$$\rho_{\pm} = (D_E/D_T)\varphi$$

$$\varphi = k^2/4$$

$$k = \left(\frac{16}{15}P_{\text{edge}}\right)^{\frac{1}{3}}$$

- Solution for $F > 0$
- $k \gg k_{\text{eq}}$
- $P_{\text{edge}} = F_{\text{edge}}/D_E$ “edge Peclet #”
 $= FL/D_E$



Comparison of scaled results from steady state (-), BCF(- - -), and KMC/SOS ($\square \circ \triangle$) for $L=25, 50, 100$, with $F=1$, $D_T=10^{12}$



Unsteady Edge Model from Atomistic Kinetics



- Evolution equations for ϕ , ρ , k

$$\partial_t \rho - D_T \nabla^2 \rho = F \quad \text{on terrace}$$

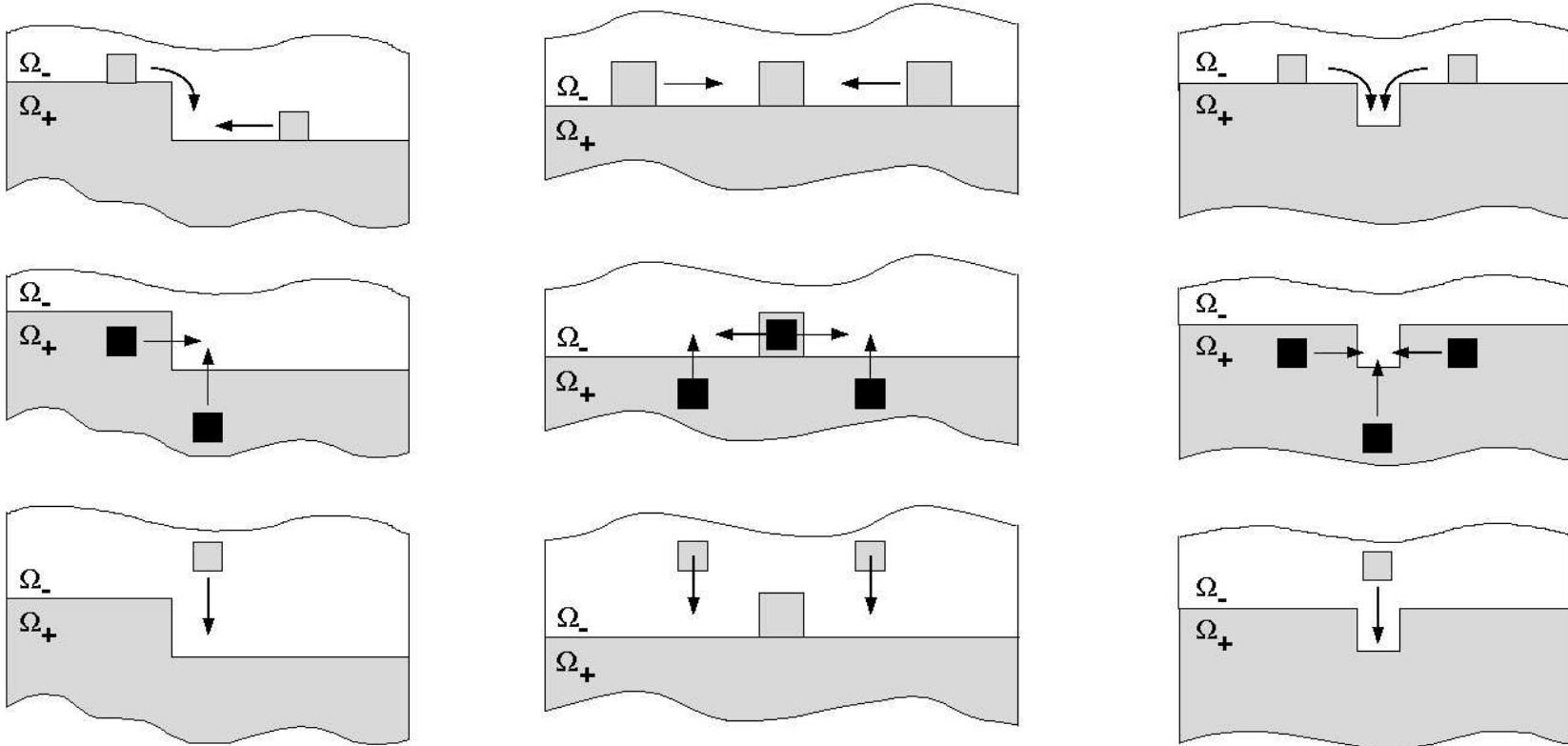
$$\partial_t \phi - D_E \partial_s^2 \phi = f_+ + f_- - f_0 \quad \text{on edge}$$

$$\partial_t k - \partial_s (w (k_r - k_\ell)) = 2 (g - h) \quad \text{on edge}$$
- Boundary conditions for ρ on edge from left (+) and right (-)
 - $v \rho_+ + D_T \mathbf{n} \cdot \nabla \rho = -f_+$
 - $v \rho_- + D_T \mathbf{n} \cdot \nabla \rho = f_-$
- Variables
 - ρ = adatom density on terrace
 - ϕ = edge atom density
 - k = kink density
- Parameters
 - D_T, D_E, D_K, D_S = diffusion coefficients for terrace, edge, kink, solid
- Interaction terms
 - v, w = velocity of kink, step edge
 - F, f_+, f_0 = flux to surface, to edge, to kinks
 - g, h = creation, annihilation of kinks



top view of step edge

Adatom and Kink Dynamics on a Step Edge



Attachment at kinks

\mathcal{M}_k kink velocity w

Kink pair creation

\mathcal{M}_k kink creation rate g

Kink pair collision

\mathcal{M}_k kink loss rate h

Reverse processes do not occur in typical MBE growth

\mathcal{M}_k no detailed balance \mathcal{M}_k nonequilibrium

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Constitutive relations



Geometric conditions for kink density

- $k_r + k_\ell = k$
 - $k_r - k_\ell = -\tan \theta$
- Velocity of step
 - $v = w k \cos \theta$
- Flux from terrace to edge,
 - $f_+ = D_T \rho_+ - D_E \varphi$
 - $f_- = D_T \rho_- - D_E \varphi$
- Flux from edge to kinks
 - $f_0 = v(\varphi \kappa + 1)$
- Microscopic equations for velocity w , creation rate g and annihilation rate h for kinks
 - $w = 2 D_E \varphi + D_T (2\rho_+ + \rho_-) - 5 D_K$
 - $g = 2 (D_E \varphi + D_T (2\rho_+ + \rho_-)) \varphi - 8 D_K k_r k_\ell$
 - $h = (2D_E \varphi + D_T (3\rho_+ + \rho_-)) k_r k_\ell - 8 D_S$



Asymptotics for Large D/F



- Assume slowly varying kinetic steady state along island boundaries
 - expansion for small “Peclet number” $f / D_E = \varepsilon^3$
 - f is flux to edge from terrace
- Distinguished scaling limit
 - $k = O(\varepsilon)$
 - $\varphi = O(\varepsilon^2)$
 - $\varphi = O(\varepsilon^2) = \text{curvature of island boundary} = X_{yy}$
 - $Y = O(\varepsilon^{1/2}) = \text{wavelength of disturbances}$
- Results at leading order
 - $v = (f_+ + f_-) + \overbrace{D_E \varphi_{yy}}^{\text{edge diffusion}}$
 - $k = c_3 v / \varphi$
 - $c_1 \varphi^2 - c_2 \varphi^{-1} v = \overbrace{(\varphi X_{yy})_y}^{\text{curvature}}$
- Linearized formula for φ
 - $\varphi = c_3 (f_+ + f_-)^{2/3} - c_4 \varphi$



Macroscopic Boundary Conditions



- Island dynamics model
 - $\rho_t - D_T \nabla^2 \rho = F$ adatom diffusion between step edges
 - $X_t = v$ velocity of step edges
- Microscopic BCs for ρ detachment

$$D_T \mathbf{n} \cdot \text{grad } \rho = D_T \overbrace{\rho - \rho_E}^{\phi} \equiv f$$
- From asymptotics
 - θ_* = reference density = $(D_E / D_T) c_1 ((f_+ + f_-) / D_E)^{2/3}$
 - γ = line tension = $c_4 D_E$
- BCs for ρ on edge from left (+) and right (-), step edge velocity

$$\begin{aligned} \pm D_T \mathbf{n} \cdot \text{grad } \rho &= D_T (\rho - \theta_*) + \gamma \kappa \\ v &= (f_+ + f_-) + c (f_+ + f_-)_{ss} + \gamma \kappa_{ss} \end{aligned}$$



Conclusions



- Kinetic model for step edge
 - kinetic steady state \neq BCF equilibrium
 - validated by comparison to SOS/KMC
- Atomistic derivation of Gibbs-Thomson
 - includes effects of edge diffusion, curvature, detachment
- Open problems
 - derivation based on distinguished limit, rather than physical regime
 - derivation of surface diffusion